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**TRANSPARENT POLYESTER FILM HAVING A HIGH OXYGEN
BARRIER AND PROCESS FOR ITS PRODUCTION**

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BACKGROUND OF THE INVENTION

The invention relates to a transparent, biaxially oriented polyester film with a base layer B, at least 80% by weight of which is composed of a thermoplastic polyester, and with at least one outer layer A. The invention further relates to the use of the film and to a process for its production.

Prior art

EP-A-0 878 297 describes a transparent, biaxially oriented polyester film with a base layer B, at least 80% by weight of which is composed of a thermoplastic polyester, and with at least one outer layer A which is composed of a mixture of polymers which contains at least 40% by weight of ethylene 2,6-naphthalate units and up to 40% by weight of ethylene terephthalate units and/or up to 60% by weight of units derived from cycloaliphatic or aromatic diols and/or dicarboxylic acids.

If the outer layer A of the film of EP-A-0 878 297 contains large amounts of ethylene 2,6-naphthalate units, the film has a tendency for delamination between the outer layer A and the base layer B. If, on the other hand, the outer layer A contains low concentrations of ethylene 2,6-naphthalate units, the thickness of this layer has to be raised in order to achieve the low oxygen permeation of not more than $80 \text{ cm}^3/(\text{m}^2 \cdot \text{bar} \cdot \text{d})$.

In a film in Example 8 of EP-A-0 878 297 the outer layer A uses pure polyethylene 2,6-naphthalate (corresponding to 100% by weight of ethylene 2,6-naphthalate units). In this case there is no significant adhesion between the outer layer A and the base layer B. The film is unsuitable for industrial use (e.g. as a composite film), since the bond releases even when subjected to a low level of mechanical stress, due to the low adhesion between the outer layer A and the base layer B of the polyester film.

In a film in Example 11 of EP-A-0 878 297, the outer layer A contains 60% by weight of ethylene 2,6-naphthalate units. In order to achieve the low oxygen

permeation demanded, below $80 \text{ cm}^3/(\text{m}^2 \cdot \text{bar} \cdot \text{d})$, the thickness of the outer layer A has to be raised to $3 \text{ }\mu\text{m}$, and this is economically disadvantageous (high capital expenditure and high material costs).

5 US-A 5,795,528 describes a coextruded film laminate which has alternating layers of PEN and PET. Like the film of EP-A-0 878 297, this film has a tendency toward delamination between the individual layers of PEN and PET. There is no significant adhesion between these layers. A laminate of this type is therefore again unsuitable for industrial use.

10 It was an object of the present invention, therefore, to provide a transparent, biaxially oriented polyester film which overcomes the disadvantage of the prior art films and in particular has improved adhesion between the individual layers. The film should be simple and cost-effective to produce, have good barrier properties, and pose no problems of disposal.

SUMMARY OF THE INVENTION

15 This object is achieved by means of a transparent, biaxially oriented polyester film with a base layer B, at least 80% by weight of which is composed of a thermoplastic polyester, and with at least one outer layer A, the characterizing features of which are regarded as being that

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- the outer layer A is composed of a copolymer or of a mixture of homopolymers and copolymers, which contains ethylene 2,6-naphthalate units in an amount in the range of from 90 to 98% by weight and up to 10% by weight of ethylene terephthalate units, and/or units derived from cycloaliphatic or aromatic diols and/or dicarboxylic acids;
 - the thickness of the outer layer A is more than $0.7 \text{ }\mu\text{m}$ and correspondingly makes up less than 25% by weight of the total film; and
 - the T_g value of the polyester film is above the T_g value of the base layer but below the T_g value of the outer layer.
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The film of the invention has low oxygen permeation, below $85 \text{ cm}^3/(\text{m}^2 \cdot \text{bar} \cdot \text{d})$, and minimum adhesion between the individual layers A and B of greater than or equal to 0.5 N/25 mm.

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DETAILED DESCRIPTION OF THE INVENTION

The structure of the film of the invention has at least two layers, and is then composed of an outer layer A and of a base layer B.

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For the purpose of this invention a polymer consisting of one type of dicarboxylic acid and one type of diol; e.g. polyethylene terephthalate, is called a „homopolymer“. If it contains a second, third or more dicarboxylic acid and/or diol it is called a „copolymer“.

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Preference is given to a polyester film in which the copolymer or the mixture of homopolymers and copolymers of the outer layer A contains ethylene 2,6-naphthalate units in a range of from 91 to 97% by weight and up to 9% by weight of ethylene terephthalate units and/or units derived from cycloaliphatic or aromatic diols and/or dicarboxylic acids. Among these, particular preference is in turn given to a polyester film in which the copolymer or the mixture of homopolymers and copolymers of the outer layer A contains ethylene 2,6-naphthalate units in a range of from 92 to 96% by weight and up to 8% by weight of ethylene terephthalate units.

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Preference is also given to a polyester film whose outer layer A has a thickness of more than $0.8 \mu\text{m}$ and correspondingly makes up less than 22% by weight of the total film, and particular preference is given to a polyester whose outer layer A has a thickness of more than $0.9 \mu\text{m}$ and correspondingly makes up less than 20% by weight of the total film.

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Examples of suitable aliphatic diols are diethylene glycol, triethylene glycol, aliphatic glycols of the formula $\text{HO}-(\text{CH}_2)_n-\text{OH}$, where n is an integer from 3 to 6 (in particular 1,3-propanediol, 1,4-butanediol, 1,5-pentanediol and 1,6-hexanediol), or branched aliphatic glycols having up to 6 carbon atoms, and cycloaliphatic diols having one or more rings and if desired containing heteroatoms. Among the cycloaliphatic diols, mention may be made of cyclohexanediols (in particular

1,4-cyclohexanediol). Examples of other suitable aromatic diols are those of the formula $\text{HO-C}_6\text{H}_4\text{-X-C}_6\text{H}_4\text{-OH}$ where X is $-\text{CH}_2-$, $-\text{C}(\text{CH}_3)_2-$, $-\text{C}(\text{CF}_3)_2-$, $-\text{O}-$, $-\text{S}-$ or $-\text{SO}_2-$. Besides these, bisphenols of the formula $\text{HO-C}_6\text{H}_4\text{-C}_6\text{H}_4\text{-OH}$ are also very suitable.

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Preferred aromatic dicarboxylic acids are benzenedicarboxylic acids, naphthalenedicarboxylic acids (for example naphthalene-1,4- or -1,6-dicarboxylic acid), biphenyl-x,x'-dicarboxylic acids (in particular biphenyl-4,4'-dicarboxylic acid), diphenylacetylene-x,x'-dicarboxylic acids (in particular diphenylacetylene-4,4'-dicarboxylic acid) or stilbene-x,x'-dicarboxylic acids. Among the cycloaliphatic dicarboxylic acids, mention may be made of cyclohexanedicarboxylic acids (in particular cyclohexane-1,4-dicarboxylic acid). Among the aliphatic dicarboxylic acids, the $\text{C}_3\text{-C}_{19}$ -alkanedioic acids are particularly suitable, where the alkane moiety may be straight-chain or branched.

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The base layer of the film is preferably composed of at least 90% by weight of the thermoplastic polyester. Polyesters suitable for this are those made from ethylene glycol and terephthalic acid (= polyethylene terephthalate, PET), from ethylene glycol and naphthalene-2,6-dicarboxylic acid (= polyethylene 2,6-naphthalate, PEN), from 1,4-bishydroxymethylcyclohexane and terephthalic acid (= poly-1,4-cyclohexanedimethylene terephthalate, PCDT), and also from ethylene glycol, naphthalene-2,6-dicarboxylic acid and biphenyl-4,4'-dicarboxylic acid (= polyethylene 2,6-naphthalate bibenzoate, PENBB). Particular preference is given to polyesters which are composed of at least 90 mol%, preferably at least 95 mol%, of ethylene glycol units and terephthalic acid units or of ethylene glycol units and naphthalene-2,6-dicarboxylic acid units. The remaining monomer units are derived from other diols and/or dicarboxylic acids. Examples of suitable diol comonomers are diethylene glycol, triethylene glycol, aliphatic glycols of the formula $\text{HO-(CH}_2)_n\text{-OH}$, where n is an integer from 3 to 6, branched aliphatic glycols having up to 6 carbon atoms, aromatic diols of the formula $\text{HO-C}_6\text{H}_4\text{-X-C}_6\text{H}_4\text{-OH}$ where X is $-\text{CH}_2-$, $-\text{C}(\text{CH}_3)_2-$, $-\text{C}(\text{CF}_3)_2-$, $-\text{O}-$, $-\text{S}-$ or $-\text{SO}_2-$, or bisphenols of the formula $\text{HO-C}_6\text{H}_4\text{-C}_6\text{H}_4\text{-OH}$ are employed.

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The dicarboxylic acid comonomer units are preferably derived from benzenedicarboxylic acids, naphthalenedicarboxylic acids, biphenyl-x,x'-dicarboxylic acids (in particular biphenyl-4,4'-dicarboxylic acid), cyclohexanedicarboxylic acids (in particular cyclohexane-1,4-dicarboxylic acid), diphenylacetylene-x,x'-dicarboxylic acids (in particular diphenylacetylene-4,4'-dicarboxylic acid), stilbene-x,x'-dicarboxylic acid or C₁-C₁₆-alkanedicarboxylic acids, where the alkane moiety may be straight-chain or branched.

The polyesters may be prepared by the transesterification process. The starting materials for this are dicarboxylic esters and diols, which are reacted using the customary transesterification catalysts, such as salts of zinc, of calcium, of lithium and of manganese. The intermediates are then polycondensed in the presence of widely used polycondensation catalysts, such as antimony trioxide or titanium salts. The preparation may be carried out just as successfully by the direct esterification process in the presence of polycondensation catalysts, starting directly from the dicarboxylic acids and the diols.

The copolymers for the outer layer A may be prepared in three different ways:

- a) In copolycondensation, terephthalic acid and naphthalene-2,6-dicarboxylic acid are placed in a reactor together with ethylene glycol, and polycondensed to give a polyester, using the customary catalysts and stabilizers. The terephthalate and naphthalate units are then randomly distributed in the polyester.
- b) Polyethylene terephthalate (PET) and polyethylene 2,6-naphthalate (PEN), in the desired ratio, are melted together and mixed, either in a reactor or preferably in a melt kneader (twin-screw kneader) or in an extruder. Immediately after the melting, transesterification reactions between the polyesters begin. Initially, block copolymers are obtained, but as reaction time increases - depending on the temperature and mixing action of the agitator - the blocks become smaller, and long reaction times give a random copolymer. However, it is not necessary and also not always advantageous

to wait until a random distribution has been achieved, since the desired properties are also obtained with a block copolymer. The resultant copolymer is then extruded from a die and granulated.

5 c) Polyethylene terephthalate (PET) and polyethylene 2,6-naphthalate (PEN) are mixed as granules in the desired ratio, and the mixture is fed to the extruder for the outer layer A. Here, the transesterification to give the copolymer takes place directly during the production of the film. This process has the advantage of being very cost-effective, and generally gives block copolymers, the block length being dependent on the extrusion temperature, the mixing action of the extruder and the residence time in the melt.

10 In a preferred embodiment of the invention, from 0.1 to 20% by weight of the polymers of the base layer B are identical with those of the outer layer A. These are either directly admixed with the base layer B during extrusion or are in any case present in the film due to addition of regrind. The proportion of these copolymers in the base layer is selected in such a way that the base layer has crystalline character.

15 In another advantageous embodiment, the film encompasses, on the side facing away from the outer layer A, another outer layer C of polyethylene terephthalate, and this layer comprises pigments.

20 The film of the invention has a high oxygen barrier and the desired good adhesion between its individual layers.

25 If the polymers/copolymers used for the outer layer A contain less than 90% of ethylene 2,6-naphthalate units and more than 10% by weight of ethylene terephthalate units along with a thickness of the outer layer A of less than 0.7 μm , although the film then has less permeability to oxygen than a standard polyester film (composed of 100% by weight of polyethylene terephthalate), its permeability is still too high for the purposes of the present invention.

If the polymers/copolymers used for the outer layer A contain more than 98% by weight of ethylene 2,6-naphthalate units (for example pure polyethylene 2,6-naphthalate), the adhesion between the outer layer A and the base layer B becomes inadequate. When subjected to mechanical stress the film tends toward delamination, which is undesirable and makes the film unusable.

A difference from the prior art is that in the film of the invention moreover the glass transition temperature T_g of the copolymer or of the mixture of homopolymers and copolymers of the outer layer A is higher than the glass transition temperature T_g of the polymers for the base layer B. The glass transition temperature T_g of the copolymers used for the outer layer A is preferably in the range from 90 to 120°C. In the DSC (differential scanning calorimetry) determination of the glass transition temperatures T_g , the transitions of the individual layers cannot be differentiated in a composite film.

Glass transitions which are determined on biaxially oriented, heat-set films in the first heating procedure (termed T_{g1} below) are, due to crystallinity and also to molecular stresses in the amorphous fraction of the specimens, relatively small in size, distributed over a wide temperature range, and shifted to higher temperatures. Because of orientation effects in particular, they are not suitable for characterizing a polymer as such. The resolution of DSC analyzers is often insufficient to detect the glass transitions in the first heating procedure (T_{g1}) of the individual layers of the film of the invention at all, the transitions being "blurred" and small, due to orientation and crystallinity.

If, however, the specimens are melted and then rapidly cooled again to below their glass transition temperature T_g (quenched), the orientation effects are eliminated. On renewed heating, glass transitions (designated T_{g2} here) are then measured which have a greater intensity and whose temperature is characteristic of the respective polymers. However, even then it is not possible to differentiate the glass transitions of the polyesters of the individual layers, since the layers mix on melting and the polyesters present therein enter into transesterification reactions with one another. It is fully sufficient, however, to compare the temperature T_{g2} of the entire coextruded films with the temperature T_{g2} of the polymer used for the

base layer B. In known films the T_g value of the polyester for the base layer is higher than that of the coextruded film, whereas the T_g value of the polyester for the outer layer is lower than that of the polyester for the base layer and also lower than that of the coextruded film. Exactly the opposite of this applies for the film of the invention. Here, the T_g value of the coextruded film is higher than that of the polyester for the base layer B but below the T_g value of the polyester for the outer layer A.

The base layer B and the outer layer(s) may also comprise customary additives, such as stabilizers and antiblocking agents. They are expediently added to the polymer or to the polymer mixture even before melting takes place. Examples of stabilizers are phosphorus compounds, such as phosphoric acid and phosphoric esters. Typical antiblocking agents (also termed pigments in this context) are inorganic and/or organic particles, for example calcium carbonate, amorphous silica, talc, magnesium carbonate, barium carbonate, calcium sulfate, barium sulfate, lithium phosphate, calcium phosphate, magnesium phosphate, aluminum oxide, LiF, the calcium, barium, zinc and manganese salts of the dicarboxylic acids used, carbon black, titanium dioxide, kaolin, crosslinked polystyrene particles and crosslinked acrylate particles.

The additives selected may also be mixtures of two or more different antiblocking agents or mixtures of antiblocking agents of the same makeup but of different particle size. The particles may be added to the individual layers in the customary concentrations, e.g. as glycolic dispersion during polycondensation or via masterbatches during extrusion. Pigment concentrations of from 0.0001 to 5% by weight have proven advantageous. A detailed description of the antiblocking agents is found, for example, in EP-A-0 602 964.

The film may be coated and/or corona- or flame-pretreated to establish other desired properties. Typical coatings are layers which promote adhesion, are antistatic, improve slip, or have release action. These additional layers may be applied to the film by in-line coating using aqueous dispersions, before transverse orientation. The film is also particularly suitable for metallizing or coating with

ceramic substances (SiO_x , Al_2O_3). Particularly good oxygen barrier values are achieved if the outer layer A is metallized or ceramic-coated.

The polyester film of the invention may preferably also comprise a second outer layer C. The structure, thickness, and chemical makeup of the second outer layer C may be selected without reference to the outer layer A already present, and the second outer layer C may also comprise the abovementioned polymers or polymer mixtures, but these do not necessarily have to be identical with the polymers for the first outer layer A. The second outer layer C may also comprise other commonly used outer layer polymers.

Between the base layer B and the outer layer A, there may also be an intermediate layer Z. The thickness of the intermediate layer is generally above $0.1\text{ }\mu\text{m}$ and is preferably in the range from 0.2 to $20\text{ }\mu\text{m}$, particularly from 0.3 to $10\text{ }\mu\text{m}$.

The thickness of the outer layer C is generally above $0.1\text{ }\mu\text{m}$, preferably in the range from 0.2 to $5\text{ }\mu\text{m}$, and particularly from 0.2 to $4\text{ }\mu\text{m}$, and the thicknesses of the outer layers may be identical or different.

The total thickness of the polyester film of the invention may vary within wide limits and depends on the application envisaged. It is from 6 to $100\text{ }\mu\text{m}$, preferably from 8 to $50\text{ }\mu\text{m}$, particularly preferably from 10 to $30\text{ }\mu\text{m}$, the proportion of the total thickness made up by the base layer B preferably being from about 40 to 95% .

The present invention also provides a process for producing this film. It encompasses

- a) producing a film having two or more layers from a base layer B and outer layer(s) A and, where appropriate, C, by coextrusion;
- b) biaxially stretching the film, and
- c) heat-setting the stretched film.

To produce the outer layer A, it is expedient to feed granules of polyethylene terephthalate and polyethylene 2,6-naphthalate directly to the extruder in the

desired mixing ratio. At about 300°C, the two materials can be melted and can be extruded. Under these conditions, transesterification reactions can occur in the extruder and during these copolymers are formed from the respective homopolymers.

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The polymers for the base layer B are expediently fed in via another extruder. Any foreign bodies or contamination which may be present can be filtered off from the polymer melt before extrusion. The melts are then extruded through a coextrusion die to give flat melt films and are layered one upon the other. The composite film is then drawn off and solidified with the aid of a chill roll and other rolls if desired.

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The biaxial stretching is generally carried out sequentially, stretching first longitudinally (i.e. in the machine direction) and then transversely (i.e. perpendicularly to the machine direction). This leads to orientation of the molecular chains within the polyester. The longitudinal stretching can be carried out with the aid of two rollers rotating at different rates corresponding to the desired stretching ratio. For the transverse stretching, use is generally made of an appropriate tenter frame.

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The temperature at which the orientation procedure is carried out can vary over a relatively wide range and depends on the properties desired in the film. In general, the longitudinal stretching is carried out at from 80 to 130°C, and the transverse stretching at from 90 to 150°C. The longitudinal stretching ratio is generally in the range from 2.5:1 to 6:1, preferably from 3:1 to 5.5:1. The transverse stretching ratio is generally in the range from 3.0:1 to 5.0:1, preferably from 3.5:1 to 4.5:1.

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During the subsequent heat-setting, the film is held for from 0.1 to 10 s at a temperature of from 150 to 250°C. The film is then wound up in a conventional manner.

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A great advantage of this process is that it is possible to feed the extruder with granules, which do not block the machine.

A further advantage of the invention is that the production costs of the film of the invention are only insignificantly greater than those of a film made from standard polyester raw materials. The other properties of the film of the invention which are relevant to its processing and its use remain essentially unchanged or have even been improved. It has also been ensured that cut material arising directly in the plant during film production can be used again in the form of regrind for film production in amounts of up to 60% by weight, preferably from 10 to 50% by weight, based in each case on the total weight of the film, without any significant resultant adverse effect on the physical properties of the film.

The film of the invention has excellent suitability for packaging food or other consumable items. The film of the invention has excellent barrier properties, in particular with respect to oxygen. It has been assured that the individual layers of the laminate remain adhering to one another when the film is processed, e.g. to give film laminates, and do not delaminate.

It is also possible to improve the gloss and the haze of the film, compared with prior art films. It has been ensured that regrind can be reintroduced to the extrusion process during production of the film in an amount of up to 60% by weight, based on the total weight of the film, without any significant resultant adverse effect on the physical properties of the film.

The excellent handling properties of the film and its very good processing properties make it particularly suitable for processing on high-speed machinery.

The table below (Table 1) gives the most important film properties of the invention again at a glance.

Table 1

	Range according to the invention	Preferred	Particularly preferred	Unit	Test method
Outer layer A					
Ethylene 2,6-naphthalate units	90 to 98	91 to 97	92 to 96	% by weight	
Ethylene terephthalate units	< 10	< 9	< 8	% by weight	
Thickness	0.7 μm to 25% of the total thickness	0.8 μm to 22% of the total thickness	0.9 μm to 20% of the total thickness	$\mu\text{m}/\%$ by weight	
Film properties					
Oxygen permeation	< 85	< 80	< 75	$\text{cm}^3/(\text{m}^2\cdot\text{bar}\cdot\text{d})$	DIN 53 380, Part 3
Adhesion between the layers	> 0.5	> 0.7	>1.0	N/25 mm	internal

Test methods

The following methods were utilized to characterize the raw materials and the films:

5 *Oxygen permeability*

The oxygen barrier test took place using a Mocon Modern Controls (USA) OX-TRAN 2/20, as in DIN 53 380, Part 3.

10 *SV (standard viscosity)*

The standard viscosity SV (DCA) is measured by analogy with DIN 53726, at 25°C in dichloroacetic acid. The intrinsic viscosity (IV) is calculated from the standard viscosity as follows

$$IV = [\eta] = 6.907 \cdot 10^{-4} \text{ SV (DCA)} + 0.063096 \text{ [dl/g]}.$$

15 *Coefficient of friction*

The coefficient of friction was determined to DIN 53 375. The coefficient of sliding friction was measured 14 days after production.

20 *Surface tension*

Surface tension was determined using what is known as the ink method (DIN 53 364).

25 *Haze*

The haze of the film was measured to ASTM-D1003-52. The Hölz haze measurement was determined by analogy with ASTM-D1003-52, but, in order to utilize the ideal measurement range, measurements were taken on four pieces of film laid one on top of the other, and a 1° slit diaphragm was used instead of a 4° pinhole.

30 *Gloss*

Gloss was determined to DIN 67 530. The reflectance was measured, this being a characteristic optical value for a film surface. Based on the standards ASTM-D523-78 and ISO 2813, the angle of incidence was set at 20° or 60°. A

beam of light at the set angle of incidence hits the flat test surface and is reflected and/or scattered thereby. A proportional electrical variable is displayed representing light rays hitting the photoelectric detector. The value measured is dimensionless and must be stated together with the angle of incidence.

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Glass transition temperatures T_g

The glass transition temperatures T_{g1} and T_{g2} were determined with the aid of DSC (differential scanning calorimetry) on film specimens. A DuPont DSC 1090 was used. The heating rate was 20 K/min, and the specimen weight was about 12 mg. The glass transition T_{g1} was determined in the first heating procedure. Many of the specimens showed an enthalpy relaxation (a peak) at the beginning of the step-like glass transition. The temperature taken as T_{g1} was that at which the step-like change in heat capacity - ignoring the enthalpy relaxation peak - achieved half of its height in the first heating procedure. In all cases, there was only a single glass transition stage in the thermogram in the first heating procedure. It is possible that the enthalpy relaxation peaks obscured the fine structure of the transition, or that the resolution of the device was not adequate to separate the small, "blurred" transitions of oriented, crystalline specimens. In order to eliminate their heat history the specimens were held at 300°C for 5 minutes after the heating procedure, and then quenched with liquid nitrogen. The temperature for the glass transition T_{g2} was taken as the temperature at which the transition reached half of its height in the thermogram for the second heating procedure.

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Adhesion between the layers

Prior to adhesive bonding, the specimen of film (300 mm long x 180 mm wide) of the present invention is placed on a smooth piece of card (200 mm long x 180 mm wide; about 400 g/m², bleached, outer laps coated). The overlapping margins of the film are folded back onto the reverse side and secured with adhesive tape.

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For adhesive bonding of the film according to the present invention, use is made of a standard polyester film of 12 µm thickness (e.g. Melinex 800), and a doctor device and doctor bar No. 3 from Erichsen, applying about 1.5 ml of adhesive (Novacote NC 275 + CA 12; mixing ratio: 4/1 + 7 parts of ethyl acetate) to the outer layer A of the film of the present invention. After aerating to remove the solvent, the

standard polyester film is laminated to outer layer A of the film of the present invention using a metal roller (width 200 mm, diameter 90 mm, weight 10 kg, to DIN EN 20 535). The lamination parameters are:

5	Amount of adhesive:	5 +/- 1 g/m ²
	Aeration after application of adhesive:	4 min +/- 15 s
	Doctor thickness (Erichsen):	3
	Doctor speed level:	about 133 mm/s
	Bond curing time:	2 h at 70°C in a circulating air drying cabinet

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A 25 ± 1 mm strip cutter is used to take specimens about 100 mm in length. About 50 mm of composite is needed here, and 50 mm of unbonded separate laps for securing/clamping the test specimen. The test specimens are secured to a sheet metal support by means of double-sided adhesive tape, by way of the entire surface of the reverse side of the film of the present invention (base layer B or outer layer C). The sheet with the composite adhesively bonded thereto is clamped into the lower clamping jaw of the tensile test machine. The clamp separation is 100 mm. The unlaminated end of the standard polyester film is clamped into the upper clamping jaw of the tensile test machine (e.g. Instron, Zwick) so that the resultant peel angle is 180°. The average peel force in N/25 mm is given, rounded to one decimal place.

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	Specimen width:	25 mm
	Pretensioning force:	0.1 N
	Test length:	25 mm
25	Separation rate until pretensioning force applied:	25 mm/min
	Start position:	5 mm
	Test displacement:	40 mm
	Sensitivity:	0.01 N
	Separation rate:	100 mm/min

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The peel force test result is equivalent to the minimum adhesion between the layers, since the adhesion between the adhesive and the standard film is markedly greater. A UV lamp, for example, can be used to demonstrate the release of the outer layer A from the base layer B of the film of the present invention. The UV light has a blueish appearance if copolymer of PEN and PET is present on the adhesive and this layer is irradiated using a UV lamp.

Examples

The following examples illustrate the invention. Information on each of the products used (trademark and manufacturer) is given only once, and this is then applicable to the examples which follow.

Example 1

Chips of polyethylene terephthalate and polyethylene 2,6-naphthalate in a mixing ratio of 3 : 97 were dried at a temperature of 160°C to residual moisture below 100 ppm, and fed directly to the extruder for the outer layer A. There, the two materials were extruded at around 300°C. The melt was filtered, shaped to a flat film in a coextrusion die and placed as outer layer A over the base layer B. The coextruded film was ejected via the die lip and solidified on a chill roll. The residence time of the two polymers in the extrusion was about 5 minutes. A copolymer was formed in the extrusion under the stated conditions.

Chips of polyethylene terephthalate were dried at a temperature of 160°C to residual moisture below 100 ppm and fed to the extruder for the base layer B. In addition, chips of polyethylene terephthalate and antiblocking agent were likewise dried at 160°C to residual moisture of 100 ppm and fed to the extruder for the outer layer C. The conditions in the extruder for the outer layer C were the same as those for the coextruder A.

Coextrusion followed by stepwise longitudinal and transverse orientation was used to produce a transparent three-layer ABC film with a total thickness of 12 µm. The thickness of the outer layer A was 1.1 µm and that of the outer layer C was 1.0 µm.

Outer layer A:

97% by weight of polyethylene 2,6-naphthalate (®Polyclear P 100 prepolymer from KOSA, Offenbach) with an SV of 600, and

3% by weight of polyethylene terephthalate with SV of 800.

5 Base layer B:

70% by weight of polyethylene 2,6-naphthalate (®Polyclear P 100 prepolymer from KOSA, Offenbach) with an SV of 600, and

100% by weight of polyethylene terephthalate (4020 from KOSA, Offenbach) with SV of 800.

Outer layer C:

10 80% by weight of polyethylene terephthalate with SV of 800, and

20% by weight of masterbatch made from 99.0% by weight of polyethylene terephthalate and 1.0% by weight of silica particles, 50% of which had an average particle size of 2.5 µm and 50% of which had an average particle size of 0.4 µm.

The individual steps of the process were:

Extrusion	Temperatures: Layer A:	300°C
	Layer B:	300°C
	Layer C:	300°C
	Take-off roll temperature	30°C
Longitudinal stretching temperature:		122°C
Longitudinal stretching ratio:		4.5:1
Transverse stretching temperature:		125°C
Transverse stretching ratio:		4.0:1
Setting temperature:		230°C

The film had the oxygen barrier required and the adhesion required between the layers.

Example 2

Coextrusion was used as in Example 1 to produce a three-layer ABC film with a total thickness of 12 μm . The thickness of the outer layer A was 1.3 μm , and the thickness of the outer layer C was 1.0 μm .

Outer layer A:

95% by weight of polyethylene 2,6-naphthalate ([®]Polyclear P 100 prepolymer from KOSA, Offenbach) with an SV of 600, and

5% by weight of polyethylene terephthalate with SV of 800.

Base layer B:

100% by weight of polyethylene terephthalate (4020 from KOSA, Offenbach) with SV of 800.

Outer layer C:

80% by weight of polyethylene terephthalate with an SV of 800, and
20% by weight of masterbatch made from 99.0% by weight of polyethylene terephthalate and 1.0% by weight of silica particles, 50% of which had an average particle size of 2.5 μm , and 50% of which had an average particle size of 1.0 μm .

The process conditions for all of the layers were as in Example 1.

Comparative Example 1c

A film was produced by analogy with Example 8 of EP-A-0 878 297. The film had the oxygen barrier required, but the adhesion between layers A and B was extremely low.

Comparative Example 2c

A film was produced by analogy with Example 1 of US-A 5,795,528, except that unlike in the example from the US-A 5,795,528 there were only 2 layers selected from PEN and PET. The film had the oxygen barrier required, but the adhesion between layers A and B was extremely low.

Table 3 gives the properties of the films produced in Examples 1 and 2 and in the Comparative Examples 1c and 2c.

Table 2

Examples	1	2	1c	2c
Ethylene 2,6-naphthalate units in the outer layer (in % by weight)	97	95	100	100
Ethylene terephthalate units in the outer layer A (in % by weight)	3	5	0	0

Table 3

Example No.	Film thickness (μm)	Layer thicknesses A/B/C (μm)	Film structure	Oxygen permeation (cm ³ /m ² bar d)	Adhesion between layers N/25 mm	Gloss (20° measurement angle) Side A Side C	Haze
1	12	1.1/9.9/1.0	ABC	78	0.6	205 175	1.8
2	12	1.3/9.7/1.0	ABC	79	1.4	199 180	1.6
1c	12	3.0/7.5/1.5	ABC	50	0.1	203 175	1.8
2c	12	6.0/6.0	AB	45	0.1	200 195	2.0